Barkhausen instabilities from labyrinthine magnetic domains

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Experimental investigations of the scaling behavior of Barkhausen avalanches in out-of-plane ferromagnetic films yield widely different results for the values of the critical exponents despite similar labyrinthine domain structures, suggesting that universality may not hold for this class of materials. Analyzing a phase-field model for magnetic reversal, we show that avalanche scaling is bounded by characteristic length scales arising from the competition between dipolar forces and exchange interactions. We compare our results with the experiments and find a good qualitative and quantitative agreement, reconciling apparent contradictions. Finally, we make some prediction, amenable to experimental verification, on the dependence of the avalanche's behavior from the film thickness and disorder.

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I. INTRODUCTION

Understanding magnetic hysteresis is a fundamental theoretical problem with important practical implications for magnetic devices.¹ Magnetic reversal is usually associated with crackling noise—the Barkhausen effect—due to the interaction between magnetic domains and structural disorder.^{2,3} The statistical properties of the Barkhausen noise are characterized by scaling laws, a signature of an underlying nonequilibrium critical behavior, suggesting that it should be possible to separate magnetic materials into well-defined universality classes characterized by the same values of the critical exponents and the same form of the scaling functions.⁴ This program has been successfully carried over for bulk materials with relatively simple parallel domain structures where it is possible to predict distinct universality classes depending on the sample microstructure (amorphous or polycrystalline).⁵

The situation is more complicated in thin films, which can show complex domain structures due to the interplay of magnetic anisotropies and dipolar interactions. In films with in-plane anisotropy, such as the MnAs films studied in Ref. 6, the domain structure crosses over from zigzag to rough as a function of the temperature and this change is reflected by a crossover in the universality class of Barkhausen avalanche statistics.⁶ The main effect of temperature in this material is to modify the strength of dipolar interactions,⁶ a relevant parameter for the critical behavior as confirmed by numerical simulations.⁷ The magnetic properties of films with out-ofplane anisotropy have been intensively investigated in recent years due to their potentially higher bit packing density.⁸ These systems typically display a labyrinthine domain structure as shown by high-resolution magneto-optical techniques⁹ or magnetic force microscopy.^{10,11} These techniques have been used to follow magnetic reversal and measure Barkhausen avalanches, reporting apparently contradictory results for the critical exponent of the avalanche statistics. Whether universality holds for these systems remains thus an intriguing open question.

Here, we use a phase-field model describing magnetization dynamics in an out-of-plane film^{12,13} to show that the scaling behavior of the avalanche size distribution is severely limited

by dipolar interactions that set a characteristic length scale to the problem. The phase-field model overcomes the limitations of the dipolar random-field Ising model,¹⁴ which is plagued by lattice effects. At the same time the model, owing to its scalar structure, is much easier to simulate than the full micromagnetic equations. Our numerical results are in good agreement with experimental data and allow the clarification the origin of the observed discrepancies between the exponents reported in the literature. This paper is organized as follows. In Sec II we illustrate the phase-field model used for our calculations. In Sec. III we present the results of our calculation including hysteresis loops and avalanche statistics. In the last section we compare our results with available experimental data and we draw some conclusions.

II. MODEL

In the model, the magnetization $\mathbf{M}(\mathbf{R})$ at position \mathbf{R} is perpendicular to the film plane and constant along its thickness d, as illustrated in the inset of Fig. 1. Therefore, to describe the domain's behavior, we adopt a two-dimensional phase-field model where a dimensionless scalar order parameter $m(\mathbf{R},t) = M_z(\mathbf{R},t)/M_s$ represents the local magnetization value on the film plane, M_s being the saturation magnetization.¹² The energy per unit thickness is given by

$$\mathcal{H}/d = \int d\mathbf{R} \bigg[V(m) + \frac{A(\nabla m)^2}{2} - \frac{\mu_0 M_s^2 d}{4\pi} \\ \times \int d\mathbf{R}' \frac{m(\mathbf{R})m(\mathbf{R}')}{|\mathbf{R} - \mathbf{R}'|^3} - \mu_0 M_s m(H_r + H_e) \bigg].$$
(1)

The first term mimics the anisotropy energy with a double well potential $V = -V_0(m^2/2 - m^4/4)$ giving the order parameter two equivalent preferential orientations ± 1 , with the constant V_0 determining the barrier height, in particular $V_0 = K_u/4$ with K_u uniaxial anisotropy constant of the material. The second term represents the exchange interaction, opposing any magnetization variation with strength A. The third term represents the long-range, nonlocal, dipolar interaction in the approximation of small film thickness; given a local value of m, it promotes the magnetization reversal in its surroundings

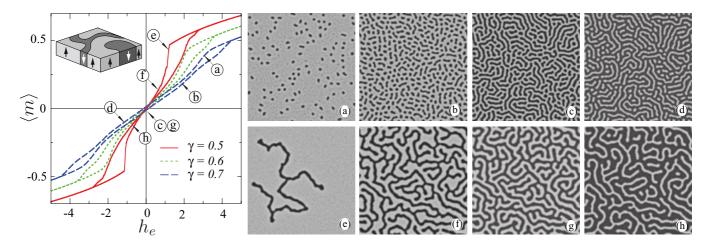


FIG. 1. (Color online) (left) Hysteresis loops for different dipolar interaction strength γ . The inset sketches our model ferromagnetic thin film. Notice that the out-of-plane magnetization is constant along the film thickness. Panels (a)–(h) represent the film magnetization m(x, y) for different external field values along the hysteresis loop for $\gamma = 0.7$ (a)–(d) and $\gamma = 0.5$ (e)–(h).

 $(\mu_0 \text{ is the vacuum permeability})$. Pinning centers are modeled by a Gaussian uncorrelated random field H_r with $\langle H_r(\mathbf{R}) \rangle = 0$ and $\langle H_r(\mathbf{R})H_r(\mathbf{R}') \rangle = \Delta \delta(\mathbf{R} - \mathbf{R}')$. Finally we consider a uniform external field H_e that is slowly ramped up and down to produce a hysteresis loop. We adopt a dimensionless scheme with unit length $\ell = \sqrt{A/V_0}$ and unit field M_s , defining $\mathbf{r} \equiv \mathbf{R}/\ell$, $h_e \equiv H_e/M_s$, $D \equiv \Delta/\ell^2 M_s^2$ and $h_r \equiv H_r/M_s$. The order parameter equation of motion can be derived in the approximation of small time fluctuations (i.e., far from the critical point) through the Hamiltonian functional derivative $\partial M(\mathbf{R},t)/\partial t = -\lambda \ \delta \mathcal{H}[M(\mathbf{R},t)]/\delta M(\mathbf{R},t)$ leading to

$$\dot{m} = \alpha \left(\frac{dV}{dm} + \nabla^2 m \right) - \gamma \int d\mathbf{r}' \frac{m(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|^3} + h_r(\mathbf{r}) + h_e(t).$$
(2)

 $\alpha = V_0/\mu_0 M_s^2$ and $\gamma = d/\ell 4\pi$ are dimensionless parameters and $\mu_0 \lambda$ is the unit time. For the numerical solution of Eq. (2) we used the finite-difference semi-implicit method of Ref. 12. All the simulations have been performed keeping $\alpha = 3$ and the disorder strength $D = 2 \times 10^{-3}$ except when another value is explicitly specified, the time step $\Delta t = 0.5$ and the squared mesh step $\Delta x = \Delta y = 0.4$.

III. RESULTS

A. Hysteresis loops and domain formation

Typical hysteresis loops obtained by a slow ramp of the external field h_e are shown in Fig. 1 together with selected plots of the domain structures (see also Supplemental Materials).¹⁵ Depending on the strength of the dipolar term, tuned by the parameter γ , the magnetization dynamics can vary considerably. For relatively large γ [see Figs. 1(a)–1(d)], we observe simultaneous nucleation of multiple droplet domains that grow as the external field is decreased below the saturation value. When the domain density is sufficiently high, domains merge giving rise to a labyrinthine structure. For smaller values of γ [see Figs. 1(e)–1(h)] a single nucleated droplet grows forming a labyrinth. This process occurs suddenly at a critical value of the external field and corresponds to a jump in the hysteresis loop. Upon further decreases in the external field, domains expand by moving domain walls. It is remarkable that while the resulting labyrinthine domain structures are similar, the dynamics that generates them is completely different. The two types of dynamics have been observed experimentally in irradiated and nonirradiated Ni films at low temperature.¹⁶ Figures 2(a) and 2(b) describe the two different domain dynamics with a color scale highlighting

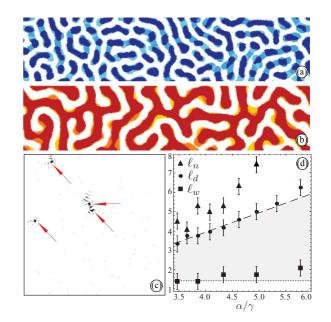


FIG. 2. (Color online) Panels (a) and (b) show the evolution of the magnetization close to the center of the loop for $\gamma = 0.7$ (a) and $\gamma = 0.5$ (b), the color scale highlights the shape of the domains at three different subsequent times. Panel (c) describes the bridging phenomenon showing the magnetization difference for two consecutive times. Four avalanches are highlighted having almost the same size, surrounded by smaller avalanches triggered by the bridging. Panel (d) shows the characteristic lengths in dimensionless units as a function of the ratio α/γ . The dashed and dotted lines are the theoretical predictions: $\ell_d = \alpha/\gamma$ for the domain width and $\ell_w = \sqrt{2}$ for the domain wall width.

subsequent avalanches. Both pictures have been taken in the central part of the hysteresis loop at the same values of the applied field for $\gamma = 0.7$ and $\gamma = 0.5$ (see also Supplemental Material).¹⁵ In Fig. 2(a), avalanches occur only close to the tails of the domain, leaving their width unchanged. When two neighboring domains are close to each other, a single avalanche bridges them together, yielding a characteristic avalanche length scale proportional to the domain width.

In Fig. 2(b) avalanches of a wide variety of sizes occur mostly by displacing domain walls. Domain bridging can be visualized by subtracting two subsequent images as shown in Fig. 2(c). There we see four bridging events having roughly the same area, together with a swarm of smaller avalanches produced by the rearrangement of the surrounding domains. This avalanche-triggering phenomenon is always present when the domain density is large.

B. Characteristic length scales

Our simulations suggest the existence of an upper bound for avalanche sizes, which appear to be limited by the width ℓ_d of the labyrinthine domains. We estimate the domain width by minimizing Eq. (1) with respect to ℓ_d assuming a simple form for the magnetization field. In particular, we consider stripe domains described by $m(x, y, \ell_d) = \sin(\pi x/\ell_d)$ and obtain $\ell_d = \alpha/\gamma$ for $H_r = H_e = 0$ and neglecting the double well potential V(m). Figure 2(d) compares the domain width estimated from simulations at $h_e = 0$ with the theoretical estimate: The agreement implies that neglecting the anisotropy in the calculation appears to be a reasonable approximation. A better estimation has been obtained in Ref. 17.

A second important length scale is given by the domain wall width ℓ_w , which should induce a lower cutoff to the avalanche distribution. If avalanche scaling is due to the dynamics of domain walls in a disordered medium then for length scales below ℓ_w the scaling should break down since the very concept of a domain wall is lost. Below this scale the inner structure of the domain wall and the rotation of the spins should become important for the avalanches, but these features can not be described by a simple scalar model. We estimate ℓ_w by energy minimization imposing a single domain wall at x = 0. In the absence of external field and neglecting dipolar interaction, Eq. (1) is minimized by a magnetization profile $m(x, y, \ell_w) =$ $tanh(x/\ell_w)$ with $\ell_w = \sqrt{2}\ell$.¹⁸ The domain wall width obtained in simulations is reported in Fig. 2(d) and compared with the theoretical estimate showing a good agreement at least for large γ .

The third relevant length scale in the problem is the diameter of the nucleation droplet ℓ_n . Bubble domains are stabilized by the dipolar interactions and ℓ_n has been estimated in ideal conditions.^{19,20} The theory, however, does not include disorder, leading to a overestimate of ℓ_n . In fact, without disorder (i.e., for D = 0) in the parameter range explored here, nucleation is completely suppressed and magnetization reversal occurs coherently (i.e., uniformly in the entire sample, without domain nucleation). (In a real sample with very little disorder, magnetization reversal is known to occur by formation of stripes (Ref. 24). They typically nucleate at the sample edges. This cannot occur in our simulations due to periodic boundary conditions thus with D = 0 the

magnetization reverses suddenly.) Hence, we conclude that the nucleation we observe is induced by disorder, but we were not able to estimate ℓ_n theoretically. In Fig. 2(d) we report the results of numerical simulations showing that ℓ_n is slightly larger than ℓ_d and grows in a similar way with α/γ . Our theoretical estimates for the characteristic length scales compare nicely with the experimental results of Im et al. on CoCrPt films9 and of Schwarz et al. on LaSrMnO₃ films.^{10,11} Using $K_u = 2 \times 10^5 \text{ J/m}^3$ and A = 3.8×10^{-12} J/m for a d = 50 nm thickness CoCrPt film,²¹ we estimate with our model $\ell_w = 12 \text{ nm}, \ell_d = 28 \text{ nm}, \text{ and}$ $\ell_n = 36 \text{ nm}$ while experiments yield $\ell_w = 15 \text{ nm}, \ell_d = 30 \text{ nm},$ and $\ell_n = 40 \text{ nm}$. For a d = 100 nm thickness LaSrMnO₃ film,¹¹ we use $K_u = 2 \times 10^4 \text{J/m}^3$ and $A = 1.7 \times 10^{-12} \text{J/m}$ and obtain $\ell_w = 26 \text{ nm}$, $\ell_d = 63 \text{ nm}$, and $\ell_n = 80 \text{ nm}$, while in experiments we have $\ell_d = 79$ nm and $\ell_n = 73$ nm, again a very good agreement.

C. Avalanche statistics

The statistical analysis of the avalanche distribution is performed exactly as in experiments.^{9,10} We resolve single avalanches by computing the difference between consecutive images of the magnetization and identifying the avalanche size S as the area of a connected cluster of reversed magnetization, defined by an appropriate threshold on the phase field (see also Supplemental Material).¹⁵ The probability distributions are obtained by logarithmic binning of the measured avalanche sizes, averaging over several realizations of the disorder field. All the measured distributions display a power-law behavior $P(S) \propto S^{-\tau}$ for at least one decade with the scaling regime limited at small and large sizes (see Fig. 3). Following Ref. 9, we first compute the avalanche distribution for different values of the applied field along the hysteresis loop as shown in Fig. 3(a) for $\gamma = 0.7$. The four displayed distributions span a region ranging from the high fields where nucleation takes place to the center of the loop where bridging events appear, passing by intermediate regions where avalanches are mainly due to domain wall motion. We notice that the distribution upper cutoff decreases as we approach the center of the loop, where the domain density is maximum. In most distributions, we observe a characteristic peak at large S that is superimposed to the apparent power-law decay. We suggest that these characteristic sizes are associated with nucleation events at small fields and bridging events at larger fields, since by removing those events we observe a marked decrease in the peaks' amplitudes. A power-law fit done excluding large values of S yields $\tau = 1.0$ in the very first stage of the nucleation process when few isolated droplets expand freely, $\tau = 1.3$ when the droplets increase their density, $\tau = 1.5$ in the intermediate region, and $\tau = 1.8$ in the loop center. While we notice that the presence of a peak could strongly bias the estimate of the exponents, the values we get are in good agreement with the experimental results of Ref. 9, which reports a similar trend in the variation of τ along the loop.

Next, we distinguish between newly nucleated droplets and the expansion of existing domains and measure the distributions of nucleated droplet sizes, which follow a Gaussian distribution [see Fig. 3(b)] with a characteristic size decreasing with γ . The areas of annihilated domains are also

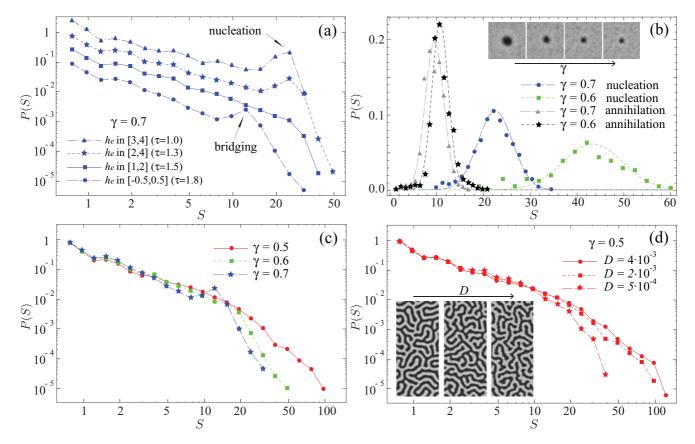


FIG. 3. (Color online) Panel (a) shows the avalanche probability distribution for $\gamma = 0.7$ measured in different regions of the hysteresis loop. The region is indicated in terms of the applied field h_e interval together with the fitted power-law exponents τ . The curves are shifted to help comparison. Logarithmic binning has been adopted. Panel (b) shows the probability distribution of the areas of nucleated droplets and for the areas observed just before domain annihilation. The inset shows the size of the very first nucleated droplet for different simulations with growing γ . Panel (c) shows the avalanche size distribution for different γ measured in the central region of the loop with h_e ranging between -0.5 and 0.5. Panel (d) shows the probability distribution of avalanche sizes for $\gamma = 0.5$ at different disorder strength D. The inset highlights the change in the domain morphology induced by the disorder strengthening.

Gaussian distributed but their typical sizes is independent of α/γ and smaller than those of nucleated droplets. This is because, contrary to nucleation, the annihilation process does not involve an energy barrier. The domain size shrinks continuously until it becomes comparable to the domain wall width, when domains become unstable and disappear. Similar observations were made experimentally in Refs. 10 and 11. In Fig. 3(c) we show the avalanche size distribution measured in the same central region of the hysteresis loop for different values of γ . The avalanche cutoff increases when γ is decreased, following the corresponding increase of the domain width and confirming that α/γ is the relevant parameter controlling the size of the scaling regime, depicted by a shadowed region in Fig. 2(d).

Finally, Fig. 3(d) shows the effect of disorder strength on the avalanche size distribution in the central part of the loop. The upper cutoff is found to increase with increasing disorder strength *D*. A similar result is found in domain wall depinning models,^{5,22} confirming that in this regime avalanches are due to domain wall motion. The inset of Fig. 3(d) refers to $\gamma =$ 0.7 and shows that, even if the domain's shape is slightly affected by the disorder strength, ℓ_d is almost independent of *D*, while the nucleation diameter ℓ_n decreases with increasing *D*. This behavior has a simple explanation: A larger value of the nucleation field H_e corresponds to stronger disorder, but higher values of H_e lead to a smaller nucleated droplet diameter ℓ_d .

IV. CONCLUSION

In this paper, we have described the process of labyrinthine domain formation using a disordered phase-field model of magnetic reversal in films with out-of-plane anisotropy. The model allows us to overcome the limitations of spin models, where lattice effects can effectively pin domain walls,¹² and at the same time provides an alternative to computationally more intense micromagnetic equations. We have modeled disorder using random fields in analogy with other wellstudied hysteresis models,²³ but our main predictions should not change substantially with other forms of disorder. To check this, we have implemented also random anisotropies¹³ finding qualitatively similar results. The model successfully reproduces the main features observed experimentally in out-of-plane magnetic thin films:⁹⁻¹¹ nucleation of nearly circular domains, domain growth and branching instabilities, labyrinthine domains expanding through avalanches, and, finally, domain annihilation. Using simple arguments we estimate the main characteristic length scales of the problem finding a good quantitative agreement with experiments. The domain wall and domain widths, ℓ_w and ℓ_d , are shown to delimit the scaling regime for the avalanches: The power-law distribution of avalanches is only observed between lower and upper cutoffs, determined by ℓ_w and ℓ_d , respectively. Such a limited scaling regime is expected to be the cause for the variability of the exponents reported in the literature for this class of materials. The competition between dipolar and exchange interactions yields the observed labyrinthine domain structure and at the same time sets strict boundaries for the scaling region.

By measuring the avalanche size distribution along the hysteresis loop, we find effective exponents that vary considerably as a function of the applied field in quantitative agreement with experimental results for CoCrPt films.⁹ We tend to attribute this variability in the exponents to a bias in the fit due to the upper and lower cutoff. The most reliable value we obtain is $\tau \simeq 1.5$ observed in the intermediate regime where nucleation and bridging events are not valid. This value is very close to the prediction of mean-field theory that is found to be accurate for bulk samples,²² although we have no theoretical argument to justify it for thin films. The values we find are, however, in net contrast with the one reported for LaSrMnO₃ film ($\tau \simeq 0.5$).^{10,11} We notice,

however, that the result is due to avalanches that occur for length scales smaller than the domain wall width ℓ_w . For this reason, they can not be captured by our model that does not involve spin rotation and other small-scale details. Yet, we have reanalyzed the data of Refs. 10 and 11 and found that on length scales larger than ℓ_w one sees a larger exponent although its determination is masked by the large-scale cutoff.

The results of our numerical simulations yield some predictions that could be tested experimentally. Figure 3(c) shows that the upper cutoff of the avalanche size distribution increases with γ . Experimentally it would be possible to tune γ by analyzing magnetic films of different thickness. Similarly, Fig. 3(d) reports a similar effect due to the strength of the disorder, which could be changed experimentally by irradiating samples, as shown in Ref. 16. Given the quantitative agreement between our model estimates and the experimentally measured length scales, we are confident that this approach should be successful.

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